

Solution properties of polystyrene in cosolvent systems

Abdel-Azim A. Abdel-Azim, Samy S. Moustafa

Mohamed M. El Dessouky, F. Abdel-Rehim† and Salah A. Hassan‡

Department of Chemistry, Military Technical College, Kobry El-Kobba, Cairo, Egypt

(Received 18 October 1985)

At 293 K, intrinsic viscosities $[\eta]$ have been measured for polystyrene samples of different relative molar mass M in mixtures of two poor solvents. These solvents were ethyl acetate and cyclohexane. Upon mixing these two poor solvents, thermodynamically better solvents could be obtained. The cosolvency was detected from the viscosity measurements. Several graphical procedures have been utilized for deriving the unperturbed dimensions of polystyrene expressed as K_θ (in the relation $[\eta] = K_\theta M^{1/2} \alpha^{1/3}$, where α is the expansion factor). It was found that the unperturbed polymer dimensions were not constant and differed from those measured in the single θ -solvent (*trans*-decalin) in which K_θ was found to be $81 \times 10^{-3} \text{ dm}^3 \text{ kg}^{-1}$.

(Keywords: solution properties; polystyrene; cosolvent systems; viscosity)

INTRODUCTION

In some previous communications¹⁻⁴ published by one of the authors of the present article, it was reported that the unperturbed dimensions of polystyrene were constant and independent on the type of solvent used. This finding was in contrast to other data published in this respect⁵⁻¹⁰. The effect of temperature has been satisfactorily^{11,12} accounted for by changes with temperature of the short-range intermolecular interactions. However, it seems that solvent effects on the unperturbed dimensions are more difficult to interpret. The effect of both polymer-solvent and solvent-solvent (in a binary solvent mixture) interactions on the unperturbed dimensions has been studied by Dondos and Benoit¹³.

The addition of a second liquid to a binary liquid polymer system to produce a ternary system is used widely for a variety of purposes. If the second liquid is a poor solvent, or a precipitant for the polymer, the dissolving potential of the liquid medium can be reduced and eventual phase separation may even occur. This does not necessarily take place in every event and sometimes mixtures of two relatively poor solvents can even produce an enhanced solvent power¹⁴. The mixed solvent is then said to exhibit a synergistic effect, which is manifested as a maximum in the limiting viscosity number $[\eta]$ curve when measured as a function of the mixed solvent composition¹⁵. Dondos and Patterson¹⁶ have shown that the sign and magnitude of the solvent-solvent interaction parameter χ_{12} could be considered as a guide to possible cosolvency. Thus, for a mixture of two nonsolvents, when χ_{12} was relatively large and positive, they might act as cosolvents for the polymer. The mixture of two solvents was unlikely to dissolve the polymer when χ_{12} was negative.

The main aim of the present investigation was to study the effect of solvent on the unperturbed dimensions of polystyrene (PS) in a series of mixed solvents exhibiting a synergistic effect. This effect was tested for the system ethyl acetate/polystyrene/cyclohexane.

EXPERIMENTAL

Materials

Ethyl acetate (EA) and cyclohexane (CH) were dried over anhydrous MgSO_4 and distilled at atmospheric pressure. The solvent mixtures were prepared by mixing appropriate volumes of pure solvents.

Five PS samples were obtained from Polymer Laboratories, Church Stretton, Shropshire, England. The quoted relative molar masses M were 1.06×10^5 , 2.94×10^5 , 4.20×10^5 , 6.40×10^5 and 9.60×10^5 for samples PS1, PS2, PS3, PS4 and PS5, respectively. The polydispersity indices \bar{M}_w/\bar{M}_n for all the polymer samples under study were ≤ 1.09 .

Techniques

Details of the preparation of binary mixtures and polymer solutions as well as measurements of $[\eta]$ have been described elsewhere^{17,18}.

RESULTS AND DISCUSSION

Results

Table 1 shows the values of $[\eta]$ for samples PS1-PS5 in EA and in binary mixtures of EA/CH having compositions expressed in terms of the volume fraction ϕ_{CH} of CH at 293 K. The data obtained in pure *trans*-decalin are also included in this table.

Mark-Houwink plots according to the equation

$$[\eta] = K_m M^a \quad (1)$$

are not reproduced here, but the derived values of the constants K_m and a also are listed in Table 1. The variation

† Also at: National Center for Radiation Research and Technology, PO Box 29, Nasr City, Cairo, Egypt.

‡ Also at: Department of Chemistry, Faculty of Science, Ain Shams University, Cairo, Egypt.

of a with the composition of the mixed solvent expressed as ϕ_{CH} is illustrated in Figure 1.

The unperturbed dimensions are normally expressed in terms of $(\langle r^2 \rangle_0/M)^{1/2}$, where $\langle r^2 \rangle_0$ is the mean square end-to-end distance in the unperturbed state. Under θ -conditions, $K_m = K_\theta$ and, therefore,

$$K_\theta = \Phi_0 (\langle r^2 \rangle_0/M)^{3/2} \quad (2)$$

where Φ_0 is the Flory constant. In *trans*-decalin, the value of K_θ could be derived directly from the Mark-Houwink plot. In the other solvents, indirect methods were applied to derive K_θ . The full equations relevant to these methods are summarized in Table 2 together with the corresponding references¹⁹⁻²³, and are designated as A, B, C, D and E. These procedures utilize various plots involving $[\eta]$ and M which allowed K_θ , for each solvent, to be derived from the intercept. The resultant values are summarized in Table 3.

Table 1 Intrinsic viscosities at 293 K for PS samples of different molar mass in EA/CH mixtures and in *trans*-decalin

| No. | Solvent ^a ϕ_{CH} | $[\eta]$ (dm ³ kg ⁻¹) | | | | | $K_m \times 10^2$ (dm ³ kg ⁻¹) | a |
|-----------------------|-------------------------------------|--|------|------|-------|-------|--|-------|
| | | PS1 | PS2 | PS3 | PS4 | PS5 | | |
| 1 | 0.00 | 30.6 | 55.5 | 68.3 | 87.4 | 110.7 | 3.60 | 0.583 |
| 2 | 0.05 | 31.6 | 58.4 | 72.3 | 93.2 | 119.0 | 2.98 | 0.602 |
| 3 | 0.10 | 33.2 | 62.3 | 77.6 | 100.7 | 129.4 | 2.60 | 0.618 |
| 4 | 0.15 | 34.8 | 66.0 | 82.6 | 107.8 | 139.5 | 2.38 | 0.630 |
| 5 | 0.20 | 36.2 | 69.7 | 87.6 | 114.8 | 148.9 | 2.15 | 0.642 |
| 6 | 0.25 | 36.9 | 71.5 | 90.1 | 118.5 | 154.1 | 2.02 | 0.649 |
| 7 | 0.30 | 37.8 | 73.7 | 93.1 | 122.6 | 159.9 | 1.93 | 0.655 |
| 8 | 0.35 | 38.9 | 75.9 | 96.0 | 126.7 | 165.9 | 1.92 | 0.658 |
| 9 | 0.40 | 39.5 | 77.8 | 98.6 | 130.4 | 170.7 | 1.82 | 0.664 |
| 10 | 0.45 | 39.7 | 78.5 | 99.5 | 131.8 | 172.6 | 1.79 | 0.666 |
| 11 | 0.50 | 39.8 | 78.7 | 99.9 | 132.3 | 173.5 | 1.75 | 0.668 |
| 12 | 0.55 | 39.5 | 78.6 | 99.4 | 131.3 | 171.8 | 1.76 | 0.668 |
| 13 | 0.60 | 39.4 | 77.7 | 98.5 | 129.9 | 169.9 | 1.83 | 0.663 |
| 14 | 0.65 | 39.2 | 77.1 | 97.7 | 129.0 | 168.8 | 1.85 | 0.662 |
| 15 | 0.70 | 38.4 | 74.9 | 94.7 | 125.0 | 163.2 | 1.89 | 0.658 |
| 16 | 0.75 | 37.5 | 72.5 | 90.0 | 119.5 | 155.5 | 2.15 | 0.645 |
| 17 | 0.80 | 36.1 | 68.5 | 85.7 | 111.8 | 144.2 | 2.49 | 0.629 |
| 18 | 0.85 | 34.5 | 63.2 | 78.3 | 101.1 | 130.0 | 3.27 | 0.601 |
| 19 | 0.90 | 31.3 | 55.9 | 68.5 | 87.0 | 109.5 | 4.38 | 0.538 |
| <i>trans</i> -decalin | | 26.4 | 43.9 | 52.5 | 64.8 | 79.4 | 8.10 | 0.500 |

^a ϕ_{CH} is the volume fraction of CH

For molar masses of samples PS1-PS5, see 'Experimental' part

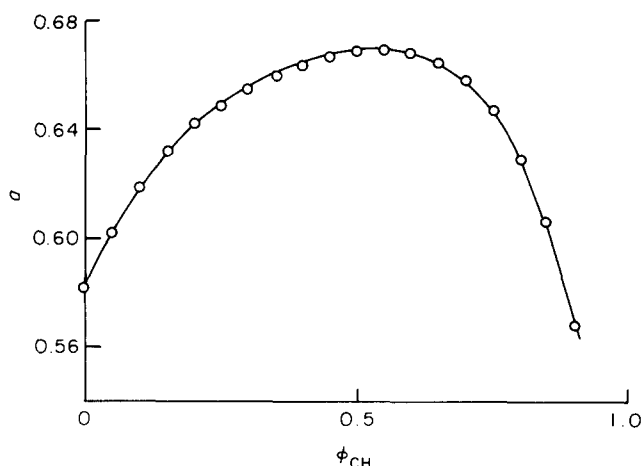


Figure 1 Dependence of a on the composition of the EA/CH binary mixtures expressed as volume fraction of CH, ϕ_{CH} , at 293 K

Table 2 Coordinates and intercept relating to extrapolation procedures used to determine K_θ

| Procedure | | Ordinate | Abscissa | Intercept on ordinate |
|-------------|------|-------------------|----------------------|---------------------------|
| Designation | Ref. | | | |
| A | 19 | $[\eta]M^{-1/2}$ | $M^{1/2}$ | K_θ |
| B | 20 | $[\eta]M^{-1/2}$ | $M^{7/20}$ | K_θ |
| C | 21 | $[\eta]M^{-1/2}$ | $M^{7/20}$ | $(\Phi_e/\Phi_0)K_\theta$ |
| D | 22 | $[\eta]M^{-1/2}$ | $M^{1/2} - \bar{D}M$ | K_θ |
| E | 23 | $\log[2K_m(1-a)]$ | $a - \frac{1}{2}$ | $\log K_\theta$ |

Table 3 Values of $10^3 K_\theta$ (in dm³ kg⁻¹) derived from different extrapolation procedures for PS in different mixed solvents 1-19 and in *trans*-decalin

| Solvent No. ^a | Method | | | | | |
|--------------------------|--------|----|-----|----|----|----|
| | A | B | C | D | E | F |
| 1 | 86 | 78 | 91 | 85 | 80 | 74 |
| 2 | 86 | 77 | 92 | 84 | 80 | 73 |
| 3 | 89 | 77 | 95 | 86 | 80 | 73 |
| 4 | 91 | 77 | 97 | 87 | 80 | 71 |
| 5 | 93 | 77 | 100 | 88 | 80 | 74 |
| 6 | 94 | 77 | 100 | 88 | 80 | 74 |
| 7 | 95 | 77 | 101 | 89 | 80 | 75 |
| 8 | 97 | 78 | 103 | 90 | 80 | 77 |
| 9 | 98 | 77 | 103 | 90 | 80 | 77 |
| 10 | 98 | 77 | 103 | 89 | 80 | 76 |
| 11 | 98 | 77 | 103 | 89 | 80 | 76 |
| 12 | 98 | 77 | 104 | 92 | 80 | 76 |
| 13 | 98 | 78 | 104 | 94 | 80 | 76 |
| 14 | 98 | 77 | 103 | 91 | 80 | 76 |
| 15 | 96 | 77 | 102 | 89 | 80 | 76 |
| 16 | 96 | 79 | 102 | 90 | 80 | 76 |
| 17 | 95 | 81 | 102 | 91 | 80 | 77 |
| 18 | 94 | 84 | 100 | 92 | 80 | 80 |
| 19 | 90 | 83 | 94 | 89 | 80 | 80 |
| <i>trans</i> -decalin | 81 | 81 | 81 | 81 | 80 | 81 |

^a For composition of solvents, see Table 1

It is to be noted that procedure E differs from the others in the respect that its plot invokes values of K_m and a for each of the solvents used. Consequently, only one value of K_θ is yielded. In this procedure, the simplified form of the original equation suggested by Abdel-Azīm and Huglin³ was used.

Another method F not involving extrapolation was employed as well. This method is considered as a semiempirical interpretation of equation (1). In this method, Munk and Halbrook²⁴ proposed the following equation:

$$K_\theta = Q^{3/(4-2a)} \quad (3)$$

where

$$Q = (K_m)(\Phi_0^{(1-2a)/3})(N_0^{1/2}M/L)^{2a-1} \quad (4)$$

In (4), (M/L) is the molar mass of polymer per unit length of chain, which was calculated by Munk and Halbrook²⁴ to be 4.14×10^9 g mol⁻¹ cm⁻¹ for PS. These authors postulated that there is no thermodynamic interaction among polymer segments within a short section of a chain with characteristic number of segments N_0 . The value of N_0 was estimated²⁴ to be ~ 9 . Invoking $\Phi_0 = 2.87 \times 10^{23}$ mol⁻¹, K_θ could be evaluated separately

for each solvent from (4). A random variation between $71 \times 10^{-3} \text{ dm}^3 \text{ kg}^{-1}$ and $80 \times 10^{-3} \text{ dm}^3 \text{ kg}^{-1}$ was obtained, leading to an average value of $76 \times 10^{-3} \text{ dm}^3 \text{ kg}^{-1}$. This method is omitted from Table 2 since an extrapolation technique to give the required intercept is not involved. Equation (3) was previously modified³ in the form

$$\log Q = [(4 - 2a)/3] \log K_\theta \quad (5)$$

Hence, the plot of $\log Q$ versus $(4 - 2a)/3$ should be linear, passing through the origin and having a slope of $\log K_\theta$. The actual plot (not reproduced here) displayed some scatter and the value of K_θ derived via least-squares analysis was $76 \times 10^{-3} \text{ dm}^3 \text{ kg}^{-1}$, exactly the same as that of the average of the individual values calculated for each solvent.

Discussion

In Figure 1, the tabulated values of exponent a are plotted as a function of the composition of the solvent. It is clear that a maximum is obtained at ϕ_{CH} between 0.5 and 0.55. This maximum was to be expected at ϕ_{CH} of 0.5 if the two solvents were of the same power. The slight shift from 0.5 seems to be due to the poverty of CH with respect to EA. Moreover, the obtained maximum illustrates clearly the synergistic effect. These findings are in good accord with the results obtained by Munk *et al.*¹⁵ for the same system.

The θ -temperature of PS in *trans*-decalin was determined experimentally by using the method of Cornet and Van Ballegooijen²⁵ and was found to be 293 K. This temperature seems to agree well with that previously found by Fukuda *et al.*²⁶ and Munk *et al.*¹⁵. Bazuaye and Huglin¹² and Inagaki *et al.*²⁷ have afforded higher values for the θ -temperature, 295.4 and 296.4 K respectively.

The plots obtained by the different methods given in Table 2 afforded good linearity in general (Figures 2–4). Some of these plots were selected in view of the fact that they exhibited a behaviour widely different from that previously obtained under other conditions³. For instance, in the Stockmayer–Fixman plot (method A), the downward deviation from linearity could be interpreted as being a result of solvent power. This deviation occurred at the highest M , but only for $a \geq 0.7$. However, in the present system, the maximum power of the solvent mixture afforded $a = 0.58$ and consequently this deviation, as expected, disappeared. Similarly, by applying method D on the same previous system³, an upward deviation was

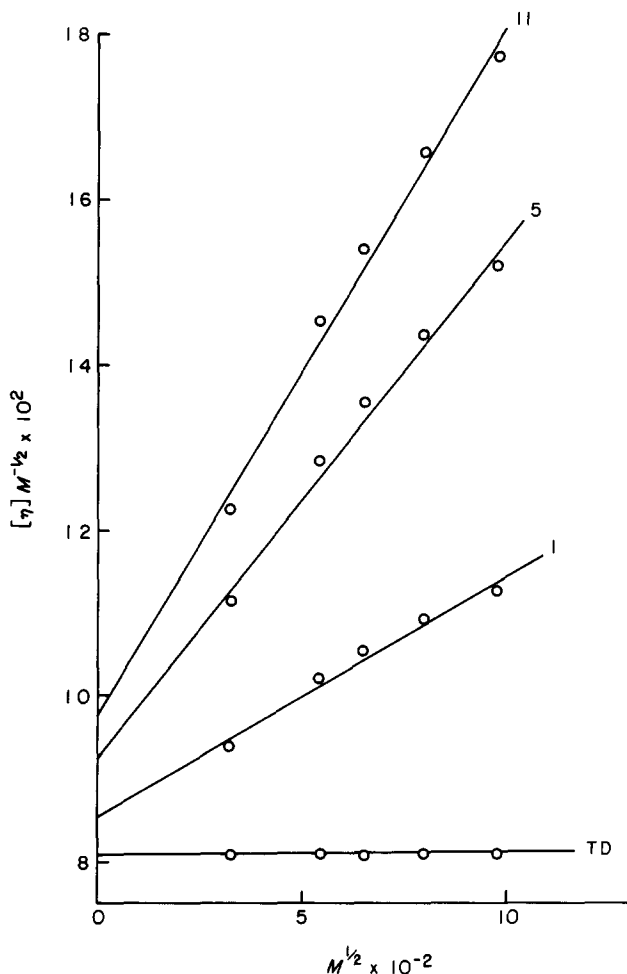


Figure 3 Stockmayer–Fixman plots (method A) for PS in *trans*-decalin (TD) and in mixed solvents 1, 5 and 11 (for compositions of mixed solvents, see Table 1)

observed at the highest M . This could be attributed³ to an excessive reduction of $M^{1/2}$ by the factor $\tilde{D}M$ when the exponent a within \tilde{D} was greater than 0.80. In the present system, a did not exceed 0.80 and hence the factor \tilde{D} was not excessively large, so that perfect linearity could be obtained.

Although the actual plots involved in methods B and C are identical (cf. Table 2), the difference in the values of K_θ in Table 3 appears to be due solely to the application of the correction factor Φ_e/Φ_0 to the intercepts obtained from method B.

The data of K_m and a , derived from measurements in all the solvent media, are embodied in a single plot (Figure 2), upon applying method E. This single plot gives good linearity. The obtained K_θ , $80 \times 10^{-3} \text{ dm}^3 \text{ kg}^{-1}$, shown in Table 3, agrees well with the value obtained in *trans*-decalin (single θ -solvent) which was found to be $81 \times 10^{-3} \text{ dm}^3 \text{ kg}^{-1}$.

In the method of Munk and Halbrook K_m and a values were also utilized but the value of K_θ obtained by applying this method (method F) was $76 \times 10^{-3} \text{ dm}^3 \text{ kg}^{-1}$ as mentioned above. This value is somewhat lower than that obtained in *trans*-decalin, which could be attributed to the uncertainty in N_0 , Φ_0 and M/L .

CONCLUSIONS

From the study of the present system EA/PS/CH and previous ones^{3,4}, we can conclude that method E seems to

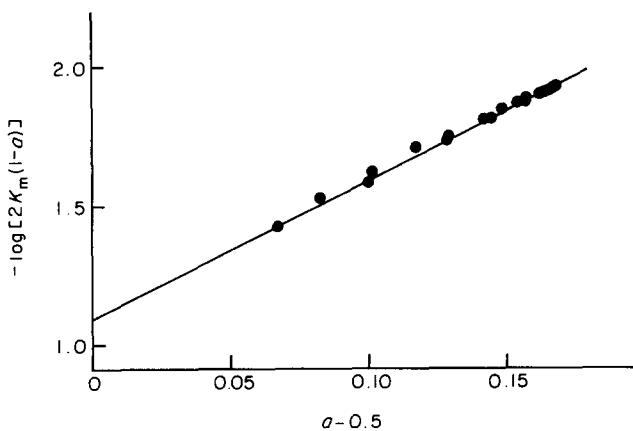


Figure 2 Plot according to method E

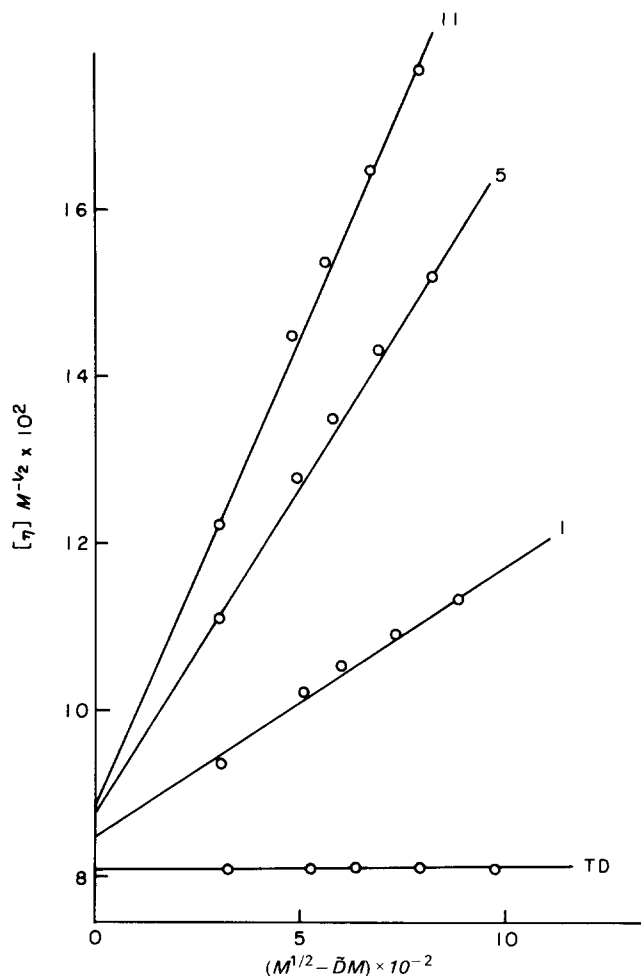


Figure 4 Plots according to method D for PS in *trans*-decalin (TD) and in mixed solvents 1, 5 and 11 (designation of solvents as in Figure 3)

be very reliable when several liquids of widely different solvent powers are available. If it is desired to use only one solvent, none of the considered methods would produce a constant value of K_θ , in contrast to the previous systems^{3,4}.

However, in view of the constancy of K_θ (and hence the unperturbed dimensions) at the particular temperature, namely 293 K in the present study, it appears that the unperturbed dimensions of polystyrene differ according to the binary solvent composition.

REFERENCES

- 1 Abdel-Azim, A.-A. A. and Huglin, M. B. *Makromol. Chem. Rapid Commun.* 1981, **2**, 119
- 2 Abdel-Azim, A.-A. A. and Huglin, M. B. *Makromol. Chem. Rapid Commun.* 1982, **3**, 437
- 3 Abdel-Azim, A.-A. A. and Huglin, M. B. *Eur. Polym. J.* 1982, **18**, 735
- 4 Abdel-Azim, A.-A. A. and Huglin, M. B. *Polymer* 1983, **24**, 1429
- 5 Vrij, A. *J. Polym. Sci. (A-1)* 1969, **7**, 1627
- 6 Dondos, A. *Makromol. Chem.* 1970, **135**, 181
- 7 Dondos, A., Rempp, P. and Benoit, H. *J. Polym. Sci. (C)* 1970, **30**, 9
- 8 Scornaux, J. and Van Leemput, R. *Makromol. Chem.* 1976, **177**, 2721
- 9 Dondos, A., Havredaki, V. and Mitsou, A. *Makromol. Chem.* 1975, **176**, 1481
- 10 Eskin, V. E., Nekrasova, T. and Juraev, U. *Eur. Polym. J.* 1975, **11**, 341
- 11 Abe, M. and Fujita, M. *J. Phys. Chem.* 1965, **69**, 3263
- 12 Bazuaye, A. and Huglin, M. B. *Polymer* 1979, **20**, 44
- 13 Dondos, A. and Benoit, H. *Macromolecules* 1971, **4**, 279
- 14 Cowie, J. M. G. and McCrindle, J. T. *Eur. Polym. J.* 1972, **8**, 1325
- 15 Munk, P., Abjaoude, M. T. and Halbrook, M. E. *J. Polym. Sci., Polym. Phys. Edn.* 1978, **16**, 105
- 16 Dondos, A. and Patterson, D. *J. Polym. Sci. (A-2)* 1969, **7**, 209
- 17 Abdel-Azim, A.-A. A. and Huglin, M. B. *Polymer* 1982, **23**, 1859
- 18 Abdel-Azim, A.-A. A. and Huglin, M. B. *Polymer* 1983, **24**, 1429
- 19 Stockmayer, W. H. and Fixman, M. *J. Polym. Sci. (C)* 1963, **1**, 137
- 20 Bohdanecký, M. *J. Polym. Sci. (B)* 1965, **3**, 201
- 21 Cowie, J. M. G. *Polymer* 1966, **7**, 487
- 22 Dondos, A. and Benoit, H. *Polymer* 1978, **19**, 523
- 23 Kamide, K. and Moor, W. R. *J. Polym. Sci. (B)* 1964, **2**, 809
- 24 Munk, P. and Halbrook, M. E. *Macromolecules* 1976, **9**, 441
- 25 Cornet, C. F. and Van Ballegooijen, H. *Polymer* 1966, **7**, 293
- 26 'Polymer Handbook', (Eds. J. Brandrup and E. H. Immergut), 2nd Edn., Wiley, New York, 1975
- 27 Inagaki, H., Suzuki, H. and Kurata, M. *J. Polym. Sci. (C)* 1966, **15**, 409